



STRUCTURAL AND MAGNETIC PROPERTIES OF α -Fe₂O₃/TiO₂ NANOCOMPOSITE DERIVED FROM LOGAS NATURAL SAND FOR ENVIRONMENTAL APPLICATION

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ABSTRACT

In this study, the iron oxide/titanium dioxide (α -Fe₂O₃/TiO₂) nanocomposites were prepared by ball milling method using Logas natural sand as raw material. The effects of milling time on the structural, magnetic properties, and chemical composition of the sample were investigated using X-ray diffraction (XRD), vibrating sample magnetometer (VSM), and X ray fluorescence (XRF) spectroscopy, respectively. The XRF measurement results found that the samples were composed of α -Fe₂O₃, TiO₂, SiO₂, and others. The crystallinity, magnetic, and composition properties of the samples were modified as milling time increases. The XRD patterns revealed that the samples have a hexagonal structure however, the crystalline size decreases with increasing milling time. The average crystallite size of iron oxide (α -Fe₂O₃), TiO₂, and α -Fe₂O₃/TiO₂ calculated based on Scherrer's formula was found to be 62.42nm, 37.12nm, and 49.77nm, respectively. The magnetic properties of the sample were studied through hysteresis loops measured using VSM to investigate the effect of milling time on the magnetic properties and found that all samples exhibited ferromagnetic behaviour with saturation magnetization (Ms) and remanence magnetization (Mr) increases to 63 and 74 %, respectively as milling time increases to 200h. The coercivity of the samples increases up to 360.10 Oe for the milling time of 100h and then decreases as the milling time increases. These coercivity values are smaller compared to that of pure hematite (α Fe₂O₃) nanoparticles. The α -Fe₂O₃/TiO₂ nanocomposites with controlled crystallite size and magnetic properties through different milling times may enable them to be used for environmental applications.

Keywords: logas natural sand, ball milling, iron oxide, titanium dioxide, and nanocomposite.

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INTRODUCTION

Metal oxides such as Fe₂O₃, Fe₃O₄, TiO₂, ZnO, CuO, SnO₂ and CeO₂ are oxides that have received a lot of attention from researchers, not only in fundamental properties but also in their technological applications ranging from high-density magnetic recording media to catalysts [1,2]. Amongst these various metal oxide catalysts, iron oxide (α -Fe₂O₃) in controllable size has received considerable attention because of its broad range of applications. Metallic magnetic iron oxide shows promising catalytic activity because it is stable and has high specific surface area, and high crystallinity [3]. Therefore, α -Fe₂O₃ nanoparticles can be an alternative nanoparticle for environmental applications and wastewater treatment [4]. Apart from being stable, iron oxide (α -Fe₂O₃) nanoparticles are abundantly available in nature [5] and have a small energy band gap, namely 2.0-2.2 eV [6], so they can absorb visible light radiation. However, the catalytic activity of α -Fe₂O₃ iron oxide depends on the particle size. Various methods have been developed to obtain iron oxide nanoparticles with controlled sizes such as co-precipitation, hydrothermal [7], sol-gel [8] microwave [9], and ball milling [10,11] methods.

However, to obtain iron oxide nanoparticles with controlled size, cheaply and efficiently, the ball milling method [12,13] can be used. Previous researchers [14,15] have used the ball milling method to prepare iron oxide nanoparticles. For example, researchers [15], prepared iron oxide nanoparticles directly from Logas natural sand-Kuansing District, Riau Province using the ball milling method and found that the ball milling product consisted of iron oxide (α -Fe₂O₃) nanoparticles with average crystallite size 39.2 nm.

The degradation efficiency of α -Fe₂O₃ iron oxide nanoparticles is reduced in water solution due to the agglomeration or large clusters due to dipolar attraction in magnetic iron oxide nanoparticles and thus reduces their specific properties of the nanoparticles. This is one of the weaknesses of α -Fe₂O₃ iron oxide nanoparticles as a catalyst. To overcome this weakness, previous researchers [16] used metal oxide nanoparticles such as titanium dioxide (TiO₂) to form a composite α -Fe₂O₃/TiO₂ with unique properties. For example, some researchers [17] recently used TiO₂/ α -Fe₂O₃ composite which was prepared using the ball milling method, and applied it as a photocatalyst to degrade dyes with high degradation



efficiency[18]. Therefore, in this research α -Fe₂O₃/TiO₂ nanocomposite prepared from Logas natural sand, Kuansing district, Riau province using the ball milling method, and their structure, magnetic properties, and chemical composition were studied so that they could be used as a catalyst for the degradation of methylene blue dye.

EXPERIMENTAL PROCEDURE

Raw Material and Chemical

α -Fe₂O₃/TiO₂ nanocomposites were prepared from natural sand from Logas village-Kuansing District Riau Province. Pure hematite (α -Fe₂O₃) nanoparticles were purchased from TokoPedia (www.tokopedia.com).

Preparation of α -Fe₂O₃/TiO₂ Nanocomposites

α -Fe₂O₃/TiO₂ nanocomposites were prepared using Logas natural sand as raw material by ball milling method as the following steps: At first, the amount of non iron oxide particles of the samples was reduced using an ironsandseparator (ISS). The obtained iron oxide particles were then processed by a strong NdFeB magnet to separate between iron oxide and non-iron oxide particles before undergoing a 4-stage ball milled process with 15 milling balls of 20 mm-diameter iron balls in the vial. The first stage of milling lasted for 50 hours and was followed by separation between iron oxide and non-iron oxide using a strong NdFeB magnet. The second, third, and fourth stages of milling were carried out for 50 hours each and followed by separation between iron oxide and non-iron oxide using a strong NdFeB magnet for each stage. To investigate the influence of milling time on the α -Fe₂O₃/TiO₂ nanocomposite, pure hematite (α -Fe₂O₃) nanoparticles were used for comparison.

Characterization of α -Fe₂O₃/TiO₂ Nanocomposites

The prepared samples were examined by XRD, VSM, and XRF to study their structural, magnetic, and compositional properties. The structural properties of the samples were examined by X-ray diffraction (XRD; PAN analytical Xpert Pro.) equipped with Cu-K α Radiation in the range of 10-100°. To analyse the magnetic properties, the samples were examined by a vibration sample magnetometer (VSM). The chemical compositions of the prepared nanocomposites were examined by X-ray fluorescence (XRF) spectroscopy. In this research, pure α -Fe₂O₃ nanoparticles were used for comparison.

RESULTS AND DISCUSSIONS

Structural Analysis

The XRD patterns of α -Fe₂O₃/TiO₂ nanocomposite are displayed in Figure-1. The nanocomposite patterns revealed 9 main diffraction peaks at $2\theta = 23.968^\circ$, 32.773° , 35.381° , 40.496° , 49.019° , 53.424° , 61.824° , 63.481° , and 70.786° for milling time of 50 h. These diffraction peaks were due to the crystal plane

of the hematite structure (102), (104), (110), (113), (024), (116), (122), (214), (300), and (119) planes. The α -Fe₂O₃/TiO₂ nanocomposite was found to have the same crystal planes of hexagonal α -Fe₂O₃[13] which is in good agreement with the JCPDS (Joint Committee on Powder Diffraction Standards) data file No. JCPDS: 33-0664[19]. However, the effect of increasing milling time shifts the diffraction peaks towards the slightly lower diffraction angle region as shown in Figure-2. An additional peak of TiO₂ confirms the formation of α -Fe₂O₃/TiO₂ nanocomposite with a high degree of crystallinity revealed by sharp and narrow diffraction peaks. It was shown in Figure-2 that with increased milling time, the increased peak width indicates reduced crystallite size. The expanded XRD patterns of α -Fe₂O₃/TiO₂ nanocomposite in the range from $2\theta = 32.4^\circ$ to 33.6° show clear a shift in the diffraction peak positions compared to that for pure α -Fe₂O₃ nanoparticles.

The average crystallite size of prepared nanocomposites was determined by observing all the maximum intensity peaks in the XRD pattern using Scherrer's formula ($D = K\lambda/\beta\cos\theta$) [20]. Here K is the Scherrer's factor (0.9), λ is the wavelength of incident X-rays, β is the FWHM and θ is the Bragg's diffraction angle for the observed peak. In addition to all diffraction peaks of α -Fe₂O₃, another diffraction peak appeared at 37.313° which corresponded to the (004) plane of TiO₂ (JCPDS file No 21-1272). Moreover, another diffraction peak at $2\theta = 25.61^\circ$ corresponds to the (111) plane of silicon. XRD results demonstrated that the crystalline TiO₂ and α -Fe₂O₃ coexisted in α -Fe₂O₃/TiO₂ nanocomposite. The average crystallite sizes for α -Fe₂O₃ nanoparticles were found to decrease as milling time increased as indicated in Figure-3. The average crystallite size of TiO₂ is 37.12 nm. The average crystallite size of α -Fe₂O₃/TiO₂ nanocomposite and TiO₂ nanoparticles is smaller than that for iron oxide α -Fe₂O₃.

The results revealed that the intensity peaks of the TiO₂ peaks appear to be quite low when the milling time increases and the peak's intensity increases slightly. Moreover, the intensity of α -Fe₂O₃ peaks increases as milling time increases. The XRD patterns indicated that the α -Fe₂O₃ nanoparticles have an average crystallite size of 62.42 nm and based on the dominant peak (104) the average crystallite size is 52.31 as shown in Table-1. The decrease in the degree of agglomeration among iron oxide (α -Fe₂O₃) nanoparticles is due to the increase in milling time. This is one of the responsible parameters to produce the smaller size of iron oxide (α -Fe₂O₃)/TiO₂ nanocomposite. The chemical composition of the samples is presented in Table-2. It can be seen that the XRF data evidence the occurrence of TiO₂. The table also indicates that the α -Fe₂O₃/TiO₂ nanocomposite is not free from impurity elements such as SiO₂.

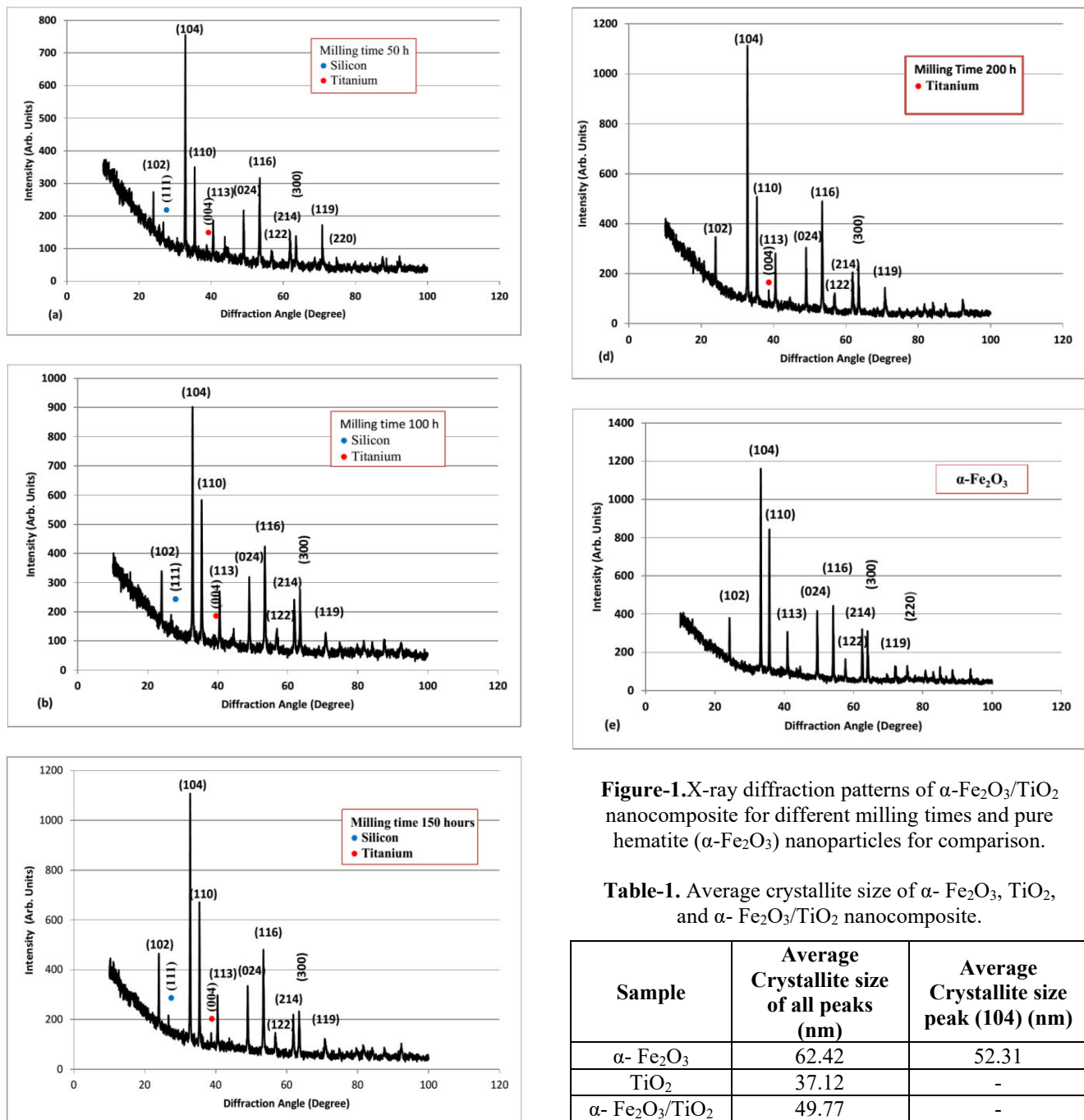


Figure-1. X-ray diffraction patterns of α -Fe₂O₃/TiO₂ nanocomposite for different milling times and pure hematite (α -Fe₂O₃) nanoparticles for comparison.

Table-1. Average crystallite size of α -Fe₂O₃, TiO₂, and α -Fe₂O₃/TiO₂ nanocomposite.

Sample	Average Crystallite size of all peaks (nm)	Average Crystallite size peak (104) (nm)
α -Fe ₂ O ₃	62.42	52.31
TiO ₂	37.12	-
α -Fe ₂ O ₃ /TiO ₂	49.77	-

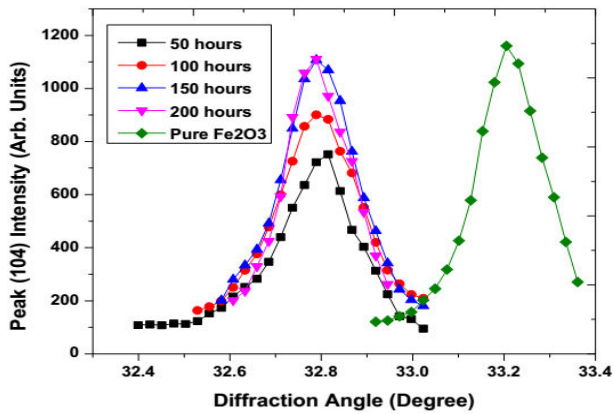


Figure-2. Expanded diffraction angle of peak (104) α -Fe₂O₃/TiO₂ nanocomposite for different milling times showing a shift of peak position to slightly lower diffraction angle and pure α -Fe₂O₃ for comparison.

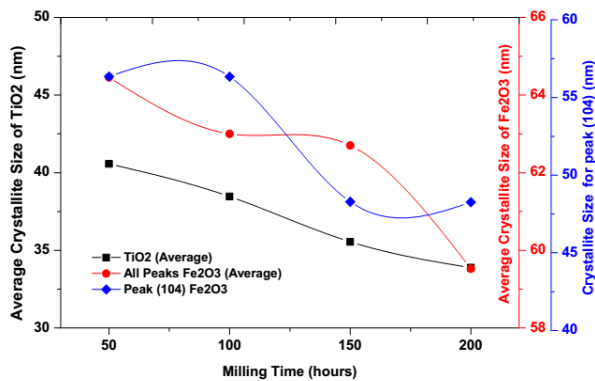


Figure-3. The average crystallite size of α -Fe₂O₃ based on all peaks, peak (104), and crystallite size of TiO₂ as a function of milling time.

Table-2. Concentration of iron oxide (α -Fe₂O₃), TiO₂, SiO₂ and other nanoparticles in the samples for different milling time.

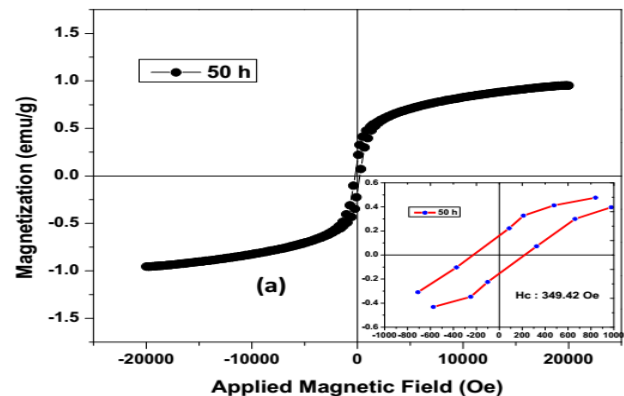
Milling Time (hours)	α -Fe ₂ O ₃ (%)	TiO ₂ (%)	SiO ₂ (%)	Others (%)
50	32.328	25.818	31.598	10.256
100	41.835	28.582	21.243	8.340
150	42.863	30.562	18.262	8.313
200	53.274	40.464	0.924	5.338

Magnetic Properties

The magnetic properties of the α -Fe₂O₃/TiO₂ nanocomposites have been studied through their hysteresis loops measured using a vibration sample magnetometer (VSM). The applied magnetic field used in this measurement sweeps from -20 to 20 kOe. As shown in Figure-4, the magnetization measurement results of all the

samples show the presence of ferromagnetic behaviour characterized by their low coercivity value. The hysteresis loops analysis shows that the saturation magnetization (M_s) and remanent magnetization (M_r) values increase with increasing the applied magnetic field and reach the saturation value under a high applied magnetic field of 20 kOe, which agrees with previous observations in earlier works[21]. The magnetic parameters consisting of M_s , M_r , and H_c of the prepared α -Fe₂O₃/TiO₂ nanocomposites are shown in Figure-5. The increase of M_s Value with increasing milling time might be because of the increasing amount of α -Fe₂O₃ which contributes to the magnetization value of the samples. The increasing amount of α -Fe₂O₃ leads to a higher magnetic moment per unit mass. The saturation magnetization of α -Fe₂O₃/TiO₂ nanocomposite is in good agreement with those reported by previous researchers[22].

The coercivity of the α -Fe₂O₃/TiO₂ prepared with a milling time of 50 h is about 349.42 Oe and increases to 360.10 Oe and then decreases to 357.94 and 88.50 Oe with increasing milling time of 100, 150 and 200 h, respectively. Thus the increase of coercivity in this range (50 to 100 h) of milling time can be qualitatively understood in terms of the effect of more segregation among oxide nanoparticles, which ultimately reduces exchange coupling between weakly coupled iron oxide nanoparticles or clusters[23]. Furthermore, magnetization data are smaller than those of pure Fe₂O₃ nanoparticles as indicated in Figure-6. The magnetic data of Fe₂O₃ nanoparticles are $M_s = 0.899$ emu/g, $H_c = 1397.71$ Oe. Comparing these data indicates that all α -Fe₂O₃/TiO₂ nanocomposites exhibit high saturation magnetization compared to that of pure α -Fe₂O₃ nanoparticles, which is related to the high magnetization of other atoms in the samples. However, α -Fe₂O₃/TiO₂ nanocomposite derived from Logas natural sand has smaller coercivity (H_c) values as compared to that of pure α -Fe₂O₃ nanoparticles, which proved their better superparamagnetic behaviour. Hence, it can be concluded that the α -Fe₂O₃/TiO₂ nanocomposite improves its superparamagnetic properties.



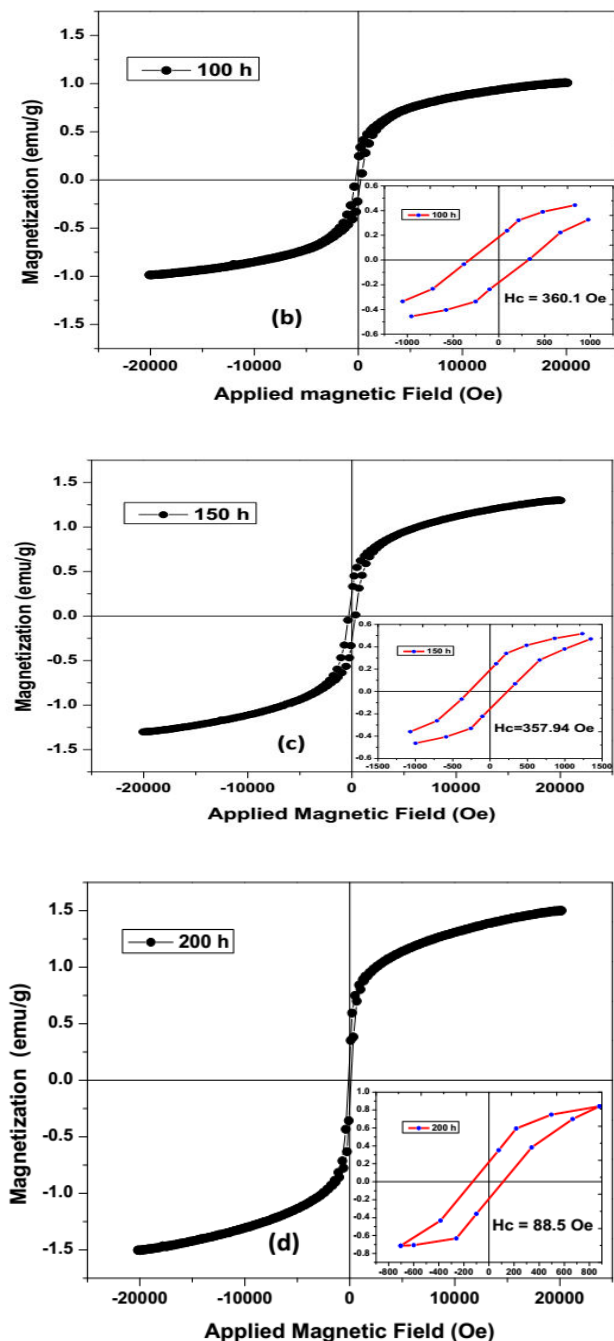


Figure-4. Hysteresis loops of $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$ nano composite with different milling times and the inset showing the enlarged loop of the samples.

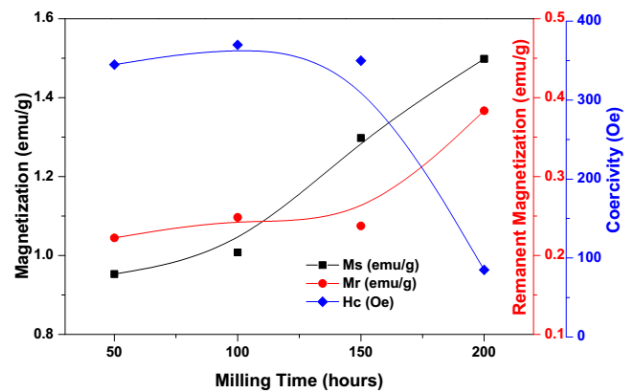


Figure-5. Saturation magnetization (M_s), remanent magnetization (M_r), and coercivity (H_c) of $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$ nanocomposite as a function of milling time.

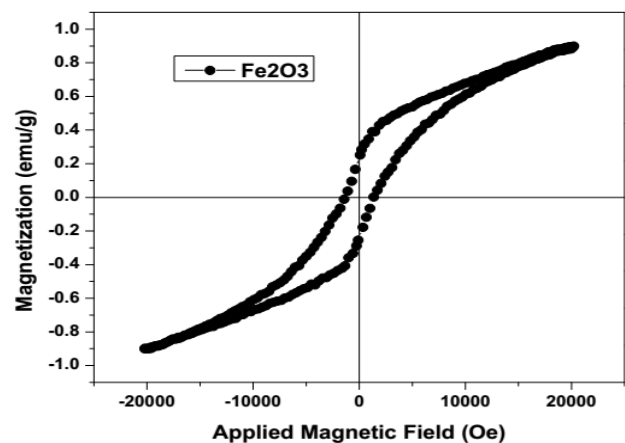


Figure-6. Hysteresis loop of pure hematite ($\alpha\text{-Fe}_2\text{O}_3$) nanoparticles.

CONCLUSIONS

In this study, $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$ nanocomposites were prepared by milling method. The XRD measurement revealed that the sample has a hematite phase with a high degree of crystalline. The measurements also confirmed the crystallite size decreases as milling time increases. The results of the VSM analysis showed that increasing milling time from 50h to 200h caused 63% and 74% increase in the saturation magnetization and remanence magnetization values, respectively. The coercivity of the samples increases with increasing milling time up to 100h and decreases rapidly after 150h. The XRF study showed that the samples are not free from impurity elements such as SiO_2 . The ability to modify $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$ nanocomposites with controlled crystallite size through different milling times may enable them to be used for environmental applications.

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REFERENCES

- [1] Frey N. A., Peng S., Cheng K., Sun S. 209. Magnetic nanoparticles: Synthesis, functionalization, and applications in bio imaging and magnetic energy storage. *Chem. Soc. Rev.* 38, pp. 2532-2542.
- [2] Zhiqin Cao, Mingli Qin, Yueru Gu, BaoruiJia, Pengqi Chen, Xuanhui Qu. 2016. Synthesis and characterization of Sn doped hematite as a visible light photocatalyst, *Mater. Res. Bull.* 77: 41-47.
- [3] B. Silke and A. Ingo. 2016. Manetic Nanocomposites. *Current opinion in Biotechnology.* 06, 39, 89-96.
- [4] Pegu R., Majumdar K. J., Talukdar D. J., Pratihari S. 2014. Oxalate capped iron nanomaterial: from methylene blue degradation to bis (indolyl) methane synthesis. *Rsc. Advances.* 4: 33446-33456.
- [5] Erwin Amiruddin, Amir Awaluddin, Meilan Sihombing, Azura Royka and Tissa Syahrul. 2020. Morphology and structural properties of undoped and cobalt doped magnetic iron oxide particles for improving the environmental quality. *International Journal of Engineering and Advanced Technology (IJEAT).* 9(6): 18-21.
- [6] Gilbert B., Frandsen C., Maxey er, Sherman D. M. 2009. Band-gap measurements of bulk and nanoscale hematite by soft X-ray spectroscopy. *Phys rev B* 79: 35108-1-35108-7
- [7] Tadica M., Panjan M., Damjanovic V., Milosevic I. 2014. Magnetic properties of hematite (α -Fe₂O₃) nanoparticles prepared by hydrothermal synthesis method. *Appl Surf Sci.* 320, 183-7.
- [8] N. Abbas, G. N. Shao, M. S. Haider, S. M. Imran, S. S. Park, H. T. Kim. 2016. Sol-gel synthesis of TiO₂-Fe₂O₃ systems: Effects of Fe₂O₃ content and their photocatalytic properties *J. Ind. Eng. Chem.* 39, 112120.
- [9] L. Hu, A. Percheron, D. Chaumont. 2011. Microwave-assisted one-step hydrothermal synthesis of pure iron oxide nanoparticles: magnetite, maghemite, and hematite. 198-205.
- [10] A. Erwin, S. Salomo, P. Adhy, N. Utari, W. Ayu, Y. Wita and S. Nani. 2020. Magnetic iron oxide particles (Fe₃O₄) are fabricated by ball milling to improve the environmental quality. *IOP Conf. Series: Materials Science and Engineering.* 845: 012051.
- [11] S. Razavi-Tousi and J. Szpunar. 2015. Effect of ball size on the steady state of aluminum powder and efficiency of impacts during milling. *Powder Technology.* 284: 149-158.
- [12] Erwin Amiruddin, Salomo Sinuraya, Amir Awaluddin, Martha Rianna, Heri Hadianito, Muhammad Rizki, Novia Magdalena Purba and Indah Tamara Sitorus. 2023. Chromium Doped Iron Oxide Nanoparticles and Physical Properties Prepared from Logas Natural Sand and Their Application in Photo-Fenton Degradation of Methylene Blue Dye. *ARPJ Journal of Engineering and Applied Sciences.* 18(10): 1102-1110.
- [13] B. Sharma, P. K. Boruah, A. Yadav and M. R. Das, 2018. TiO₂-Fe₂O₃ nanocomposite heterojunction for superior charge separation and the photocatalytic inactivation of pathogenic bacteria in water under direct sunlight irradiation, *J. Environ. Chem. Eng.*, 6(1): 134-145, doi: 10.1016/j.jece.2017.11.025.
- [14] Erwin Amiruddin, Amir Awaluddin, Innike Hariani, Ribka Sihombing and Riska Angraini. 2020. The Influence of Milling Ball Size on the Structural, Morphological, and Catalytic Properties of Magnetite (Fe₃O₄) Nanoparticles toward Methylene Blue Degradation. *Journal of Physics: Conference Series* 1655. 012006 doi:10.1088/1742-6596/1655/1/012006
- [15] Heri Hadianito, Erwin Amiruddin, Rebi Septiawan, Putri Venera and Vivi Aprilia. 2020. Structural and Morphological Properties of Undoped and Manganese Doped Hematite Nanoparticles Prepared by Ball Milling Method. *Journal of Physics: Conference Series* 1655. 012013 doi:10.1088/1742-6596/1655/1/012013
- [16] Baldrian P., Merhautova V., Gabriel J., Nerud F., Stopka P. 2006. Decolorization of synthetic dyes by hydrogen peroxide with heterogeneous catalysis by mixed iron oxides. *Appl. Catal. B* 66: 258-264.



- [17] S. Lubis, I. Maulana and M. 2018. Synthesis and Characterization of $\text{TiO}_2/\alpha\text{-Fe}_2\text{O}_3$ Composite using Hematite from Iron Sand for Photodegradation Removal of Dye. J. Nat. 18(1): 38-43, doi: 10.24815/jn.v18i1.8649.
- [18] S. C. Lee, H. O. Lintang and L. Yuliati. 2017. High photocatalytic activity of $\text{Fe}_2\text{O}_3/\text{TiO}_2$ nanocomposites prepared by photodeposition for degradation of 2, 4-dichlorophenoxyacetic acid. Beilstein J. Nanotechnol., 8(1): 915-926, doi: 10.3762/bjnano.8.93.
- [19] Xuehui Rao, Xintai Su, Chao Yang, Jide Wang, Xiping Zhen, and Daishun Ling. 2013. From spindle-like $\beta\text{-FeOOH}$ nanoparticles to $\alpha\text{-Fe}_2\text{O}_3$ polyhedral crystals: shape evolution, growth mechanism, and gas sensing property. Cryst Eng Comm. 15, 7250.
- [20] B. D. Cullity. 1978. Elements of X-Ray Diffraction (Addison-Wesley, Boston). p. 102.
- [21] Erwin Amiruddin, Salomo Sinuraya, Rahmondia N. Septiadi, Yanuar Hamzah, Amir Awaluddin, Loly A. Hardani, Fitri A. Lestari, Yessi Magdalena and Devi T. Gurning. 2022. A Systematic Study of the Structural and Magnetic Properties of Nickel Doped $\alpha\text{-Fe}_2\text{O}_3$ Nanoparticles Prepared from Logas Natural Sand. ARPJ Journal of Engineering and Applied Sciences. 17(14): 1408-1413.
- [22] Houda Mansour, K. Omri, Radhouane Bargougui, Salah Ammar. 2020. Novel $\alpha\text{-Fe}_2\text{O}_3/\text{TiO}_2$ nanocomposites with enhanced photocatalytic activity. Applied Physics A. 126:151 <https://doi.org/10.1007/s00339-020-3320-3>
- [23] Erwin and Adhy Pryayitno. 2017. Magnetic exchange interaction in cobalt samarium thin films for high-density magnetic recording media. ARPJ Journal of Engineering and Applied Sciences. 12, 12, 3832-3835.